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R. Laine, Y. Blum, A. Chow, a	and K. Schwartz					
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The program objectives are						
to designed, tractable silicon of SRI developed technology; (2) Mo						
preceramic fibers; and, (3) Pyro	olvsis technique	es for trans	forming the	preceramic	fibers into	
high strength Si ₂ N, and silicon	carbide nitride	(SiCN) fib	ers. In the	e past year	, we have	
learned to prepare polysilazane	s drived from pr	ecursors of	the type-[H SiNMe] 📑,	whose vis-	
coelastic properties can be care	efully controlle	ed by type o	of catalyst	and/or reac	tion condi-	
tions. This control has permit 10 µm as seen in the attached p	ted us to draw p	receramic i	ilbers of di	ameters as loned pyrol	smarr as vsis methodo:	
logy that permits us to obtain	ceramic vields o	of 50-70% wi	th Si N, pu	rities rang	ing from 80-	
99%. We have discovered that pe	olymer molecular	weight gre	atly influe	nces the ce	ramic yield	
but only monomer design or pyro.	lysis under a re	eactive atmo	sphere seem	s to influe	nce selecti-	
vity to specific ceramic produc	ts as shown in t	he attached	l Table. 🔑			
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Technical Report No. 8

SILICON NITRIDE CERAMIC FIBERS FROM PRECERAMIC POLYMERS

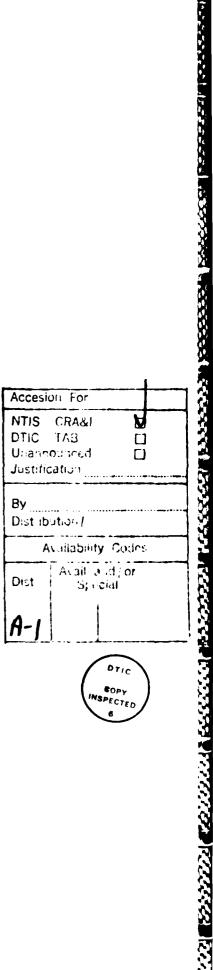
Richard M. Laine, Yigal D. Blum, Andrea Chow, and Kenneth S. Schwartz Inorganic and Organometallic Chemistry, Physical Polymer Chemistry Program and the Ceramics Program SRI International, Menlo Park, CA 94025

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POLYMER PRECURSORS TO SILICON NITRIDE COATINGS, BINDERS AND FIBERS

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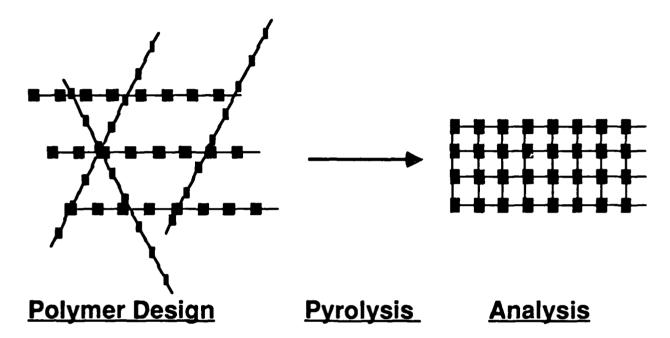
Work Sponsored By SDIO/IST As Managed By The Office
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PROGRAM OBJECTIVES

- Synthesize tractable polymer precursors to silicon nitride, using SRI's catalytic dehydrocoupling process nitride that can be spun and that give high ceramic yields of high purity Si₃N₄.
- Develop an understanding of the kinetics and mechanisms of the catalytic process.
- Detail the conditions necessary to shape the polymer precursor into a finished, infusible preceramic form.
- Detail the pyrolysis conditions necessary to transfrom the infusible shape into a finished, high density ceramic product.
- Develop analytical methods of characterizing the final ceramic product.

Polymer precursor design, synthesis and pyrolytic transformation make up the three steps in the development of preceramics useful for the preparation of coatings and fibers or for binder applications.



MOLECULAR ANALOGS OF MATERIALS

IN THEORY, Given the Empirical Formula for a Material,
It Should be Possible to Prepare a Chemical
Analog

CERAMIC MONOMERIC UNIT

CHEMICAL MONOMERIC UNIT

Si₃N₄

H₆Si₃N₄

THIS ANALOG REPRESENTS A POTENTIAL PRECURSOR TO THAT MATERIAL

MONOMERS ARE OFTEN VOLATILE and Therefore not Suitable as Precursors to Ceramics--One Needs Oligomeric or Polymeric Species

LINEAR OLIGOMERS AND POLYMERS MUST RETAIN LATENT REACTIVITY -- So They Can Be Made Infusible By Crosslinking:

POLYSILAZANE PRECURSORS TO Si₃N₄

IN PRACTICE:

It is Difficult to Synthesize Even Simple, High Molecular Weight Preceramic Polysilazanes That Are Tractable; Yet Retain Latent Reactivity.

The Polysilazane, H-[Me₂SiNH]_X-H, a Nitrogen Analog of Polysiloxane Exhibits no Latent Reactivity and Therefore Depolymerizes when Pyrolyzed --Giving No Ceramic Product

Polysilazane Syntheses by Catalytic Dehydrocoupling

SRI has recently developed a catalytic method of forming Si-N bonds from Si-H and N-H bonds that can be used to form polysilazanes:

$$Et_2SiH_2 + NH_3 \frac{Ru_3(CO)}{12^{60^{\circ}C}} + H_2 + -[Et_2Si-NH]_X - M_n \approx 500$$

The products obtained by this reaction are mostly cyclomeric. However, by performing modeling studies on this type of reaction, we been able to obtain sufficient kinetic information to establish a preliminary picture of the reaction mechanism and use this to develop better approaches to preceramic polymers as shown in the next slides:

MODELING THE DEHYDROCOUPLING REACTION

Rate =
$$k[Et_3SiH][RNH_2]^{-1.X}$$
 for R = n-Pr, n-Bu

Rate =
$$k[Et_3SiH][RNH_2]^{0.X}$$
 for R = s-Bu

Rate =
$$k[Et_3SiH]^{0.Y}[RNH_2]^{0.X}$$
 for R = t-Bu

Rate = 0 for piperidine

Rate =
$$k[Ru_3(CO)_{12}]^{-0.X}$$

PROPOSED DEHYDROCOUPLING MECHANISM

$$RNH_2 + M < \longrightarrow (RNH_2)M + RNH_2 < \longrightarrow (RNH_2)_2M$$

$$(RNH_2)M + Et_3SiH \leftarrow Et_3SiM(H)(RNH_2)$$

$$Et_3SiM(H)(RNH_2) + RNH_2 \longrightarrow Et_3SiNHR + (RNH_2)MH_2$$

$$(RNH_2)MH_2 \longrightarrow (RNH_2)M + H_2$$

The above results indicate that transition metal catalyzed dehydrocoupling is extremely susceptible to steric inhibition. This is supported by the phenyl silane coupling reactions wherein, the 60°C reaction leads exclusively to linear oligomers and only at 90°C does crosslinking occur by formation of imino bridges. This latter observation suggests that imino bridge formation could be the mechanism whereby linear preceramic polysilazanes can be made infusible.

$$PhSiH_{3} + NH_{3} \frac{Ru_{3}(CO)}{12^{60} C/THF} + H_{2} + H_{2}(CO) + H_{3}(CO) + H_{3}(CO) + H_{4}(CO) + H_{5}(CO) + H_{5}($$

$$H-[PhSiHNH]_{x}-H + NH_{3} + \frac{Ru_{3}(CO)}{12^{/90^{\circ}C}} + H_{2} + H_{3} +$$

$$^{\rm NH}_{0.5}$$

$$-{\rm [PhSiHNH]}_{\rm x}{\rm [PhSiNH]}_{\rm y} -$$

$${\rm solid, M_n = 1400}$$

Preceramic Polysilazanes

The ammonolysis of H_2SiCl_2 gives oligomers, $-[H_2SiNMe]_X$ -, where $x \approx 10$:

$$H_2SiCl_2 + 3MeNH_2 \longrightarrow -[H_2SiNMe]_x - + 2xMeNH_3Cl$$

Pyrolysis of these oligomers gives a 38-39 wt % yield of ceramic product that is reported to be mostly silicon nitride. Considerable precursor volatilization occurs during pyrolysis.

Seyferth and Wiseman, 1984

Because the Polysilazane HNMe- $[H_2SiNMe]_x$ -H has N-H caps that can react, when heated at 60-90°C, with the internal H_2Si groups. The Dehydrocoupling Reaction can be used to form tractable higher molecular weight (less volatile) polymers and then to crosslink (thermoset) these polymers to render them infusible:

$$HNMe-[H_2SiNMe]_x-H \frac{Ru_3(CO)}{12^{\frac{1}{60}}} = polymers$$

Figure 1, following, illustrates the changes in molecular weight and dispersion that occur as polymerization proceeds.

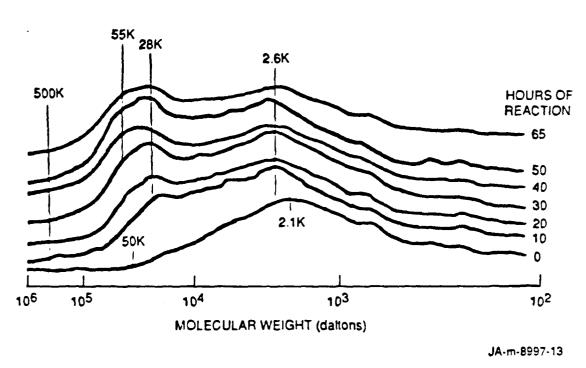
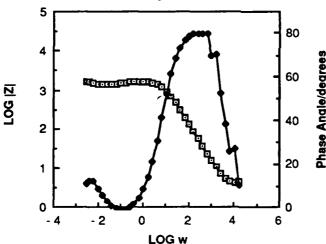


FIGURE 1 GPC RESULTS OF [H2SINMe]x POLYMERIZATION CATALYZED BY Ru3(CO)12

The tractable polymers shown in the last slide have been used to make 2000 A coatings of silicon nitride on stainless steel, aluminum, silica and graphite/graphite composites. The coatings on stainless steel were featureless in the SEM at the highest magnification. Electrochemical corrosion studies were conducted on coated aluminum 6061 coupons to establish the microporosity of these coatings and their ability to protect the surface from corrosive environments over extended periods. The results of these electrochemical corrosion studies (shown on the following slides) reveal that the coatings contained some flaws but were unchanged upon exposure to 3.5% NaCl solution for as long as 21 days. These results suggest that silicon nitride coatings prepared by simple dipcoating techniques may be useful in a variety of composite applications [e.g. as protective coatings on graphite fibers during the fabrication of aluminum/graphite fiber composites.

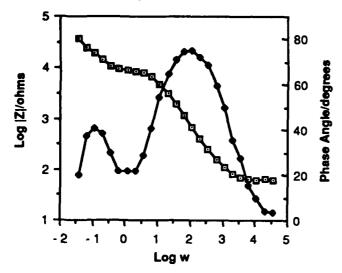
6061 Aluminum Alloy in 3.5% NaCl for 3 hours



AC Impedance Spectrum

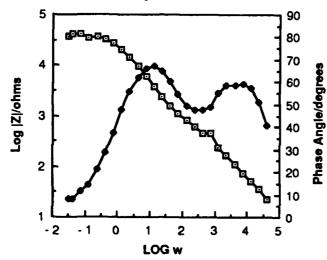
<u>Observation</u>: The Polarization Resistance R_p is at least 10^3 which is relatively low.

Coated 6061 Alloy in 3.5% NaCl for 5 hours



AC Impedance Spectrum Observation: R_p is at least $10^{4.5} = 31600$ ohms which is relatively high.

Coated 6061 Alloy in 3.5% NaCl for 21 days

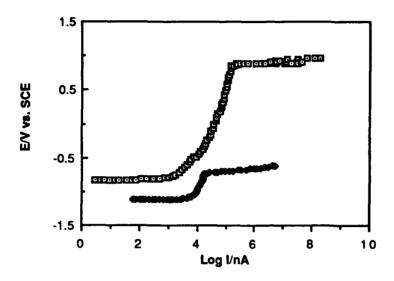


AC Impedance Spectrum <u>Observation</u>: R_p is still at least $10^{4.5}$ or 31600 ohms after 21 days in solution.

i_{corr} ≤22.6μA/cm² Uncoated 6061 T6 i_{corr} ≤2.6μA/cm² Coated 6061 T6 <u>Padazaani esampada findantsi pasamanan Parkaana Pakanan </u>

Significance: The Corrosion Protectic n Afforded by Si₃N₄ Coatings Does not Degrade with Time over 21 Days

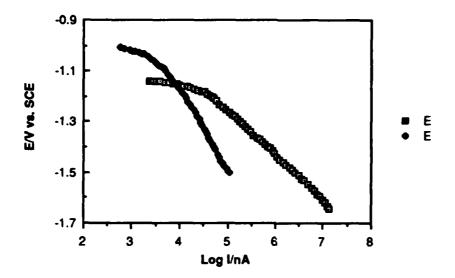
ANODIC POLARIZATION DIAGRAM



Observation: Si₃N₄ Coated Al 6061 T6 Alloy has a Pitting Potential about 1.75 Volts Higher than Uncoated alloy.

<u>Significance</u>: The coated material is significantly more resistant to pitting corrosion.

CATHODIC POLARIZATION DIAGRAM



Observation: $i_{corr} = 7.4 \,\mu\text{A/cm}^2$ Uncoated 6061 T6 Alloy $i_{corr} = 1.4 \,\mu\text{A/cm}^2$ Coated 6061 T6 Alloy by Tafel Extrapolation

<u>Observation</u>: b_c = 175 mv/decade Uncoated 6061 T6 Alloy = 340 mv/decade Coated 6061 T6 Alloy

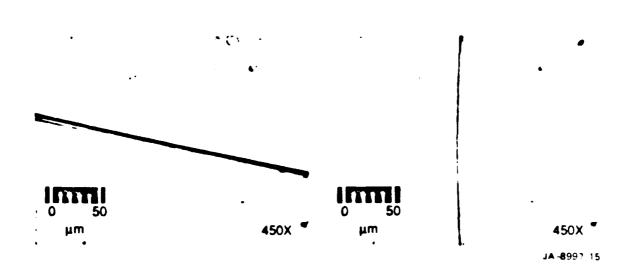
Interpretation: Based on Geometric Surface Area, the Substrate
Corrosion Rates and Hydrogen Evolution Rates are
Significantly Lower on the Coated Specimen.

These precursor polysilazane polymers have proved extremely useful as binders in the fabrication of fully dense silicon nitride bodies by pressureless sintering of compression molded silicon nitride powder. We find that pyrolysis of shapes compression molded with polysilazane at 800°C leads to densification and observable intrinsic strength in the green body. Densities of up to 75% have been obtained. Further heating at 1725°C for 20h under N₂ leads to full densification if sintering aids are present.

By comparsion, 800°C pyrolysis of shapes molded with a standard organic binder does not result in densification and the resulting product is similar to chalk.

The more viscous polymer can be extruded to give fibers of $100-300\mu m$ and hand drawn, as shown in the following Figure, to give smooth precursor fibers of approximately $10 \mu m$.

HAND DRAWN 10 µm FIBERS USING POLYMER DERIVED FROM -[H2SINMe]x=



Pyrolysis of -[H₂SiNMe]_X-Oligomers and Polymers

Oligomer	Mn	<u>Viscosity</u>	Ceramic Yield	<u>Si₃N₄</u>
	(GPC)	(poise)	(900°C)	%
[H ₂ SiNMe] _x x = 10	650	1	40	80-85
[H ₂ SiNMe] _x x = 19	1150	5	45-50	80-85
[H ₂ SINMe] _x Ru ₃ (CO) ₁₂ /90°C		18	60-65	80-85
[H ₂ SiNMe] _X Ru ₃ (CO) ₁₂ /90°C		100	65-70	80-85

The salient features seen in the above table are that there is a direct correlation between the molecular weight $M_{\rm n}$ of the precursor and the ceramic yield. This is to be expected if precursor volatilization results in physical loss of precursor. Also of importance is the fact that the viscosity of the polymer changes from 1 to 100 poises while the $M_{\rm n}$ changes from only

650 to 2300 D. These observations indicate that the polymerization process is most likely a gelation process. The mostly linear macromolecular structure of the 650 D material is therefor quite different from the highly branched species present in the 2300 D polymer. Of considerable importance is that the selectivity to ceramic products (83% Si₃N₄ and 17% amorphous carbon) remains unchanged despite the considerable change in polymer molecular weight and macroscopic structure. These results suggest that it is the monomer unit -H₂SiNMe-, at the molecular level, that determines the selectivity to ceramic products.

This last conclusion, if valid, supports the concept that it is indeed feasible to design materials at the molecular level. Finally, transition metal catalyzed dehydrocoupling has now been domonstrated for the formation of oligo-borazines from BH₃ and MeNH₂. In addition, we have found a simple condensation process that leads to polyiminotitanides which can be used as perceramic precursors to titanium nitride.

BN OLIGOMER PRECURSORS BY CATALYTIC DEHYDROCOUPLING

Pyrolysis at 800°C: Ceramic Yield 60 Wt % Pyrolysis at 1600°C: Ceramic Yield 49 Wt %

TITANIUM NITRIDE PRECURSORS

ዘመዘው የአብሮ ለአብሮ ለመብረት እንደነገር እንደነገር

Pyrolysis at 800°C under NH₃ gives TiN

PROGRESS

- Developed an understanding of the mechanism(s) of the dehydrocoupling reaction.
- Learned how to polymerize oligomers of -[H₂SiNMe]_X- and to control product rheological properties.
- Demonstrated the feasibility of preparing thin, corrosion resistant coatings on metals and silica using a preceramic polymer.
- Demonstrated the utility of using preceramic polymers as binders for compression molded Si₃N₄.
- Prepared 10-100 μm preceramic fibers from preceramic polymers.
- 6 publications in press or in preparation. 2 patent applications. One major spin-off project to improve the strength of glass bottles.

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